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Journal of Luminescence ■ (■■■■) ■■■-■■■

JOURNAL OF
LUMINESCENCEwww.elsevier.com/locate/jlumin

Femtosecond spectroscopic studies of photoinduced electron transfer in MDMO-PPV:ZnO hybrid bulk heterojunctions

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Abstract

The photophysics of charge carriers (polaron) in MDMO-PPV:ZnO hybrid bulk heterojunction is studied at 80 K by femtosecond transient absorption spectroscopy. A short-lived positive polaron is observed in the blend phase in MDMO-PPV:ZnO blend films with a weight ratio of 1:1 and 1:2. Further increase of ZnO weight ratio results in a significant quenching of the polaron absorption. The results are discussed in the concept that both pristine polymer and MDMO-PPV:ZnO blend phases coexist in the blend films. It is concluded that a polaron is photogenerated within the excitation laser pulse (<100 fs) and electron transfer efficiency is highest in blend films 1:1 and 1:2. Lack of the interfacial area and faster back electron transfer process are discussed to be responsible for the quenching of the electron transfer efficiency in blend film 1:3.

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Keywords: Ultrafast dynamics; Photophysics of polymers; Spectroscopy; Excited state dynamics of molecules; Electron transfer

1. Introduction

Organic-inorganic solar cells have been attracting considerable interest [1,2]. One of the main challenges associated with solar cells made from conjugated polymers and a nanocrystalline inorganic semiconductor is the fabrication of bulk heterojunction (BHJ) films. A new approach to make such a BHJ was reported recently in which a continuous interpenetrating network of semiconductor is created inside a thin conjugated polymer film using a TiO₂ precursor [3]. The characteristic distance between the polymer and semiconductor is in the order of 20–30 nm and results in the formation of photoinduced charges that are collected in a photovoltaic device. Photophysics of the charge carriers is vital in understanding the photovoltaic process in polymer-based devices and relevant research has proliferated in recent years [4,5].

In this paper we report femtosecond spectroscopic studies at 80 K on photoinduced polarons in MDMO-PPV:ZnO hybrid BHJ made by a similar pre-

cursor approach as the MDMO-PPV:TiO₂ BHJ cells that were mentioned above. The aim is to investigate the influence of the ratio of ZnO relative to MDMO-PPV on the polaron generation, transport and recombination processes in the blend.

2. Experimental

The preparation of MDMO-PPV:ZnO blend films via the precursor route is described in detail in Ref. [3]. The femtosecond transient absorption setup covers the spectral range from 0.9 up to 3.2 eV. Briefly, an amplified Ti:sapphire laser system equipped with two optical parametric amplifiers (OPA) was used to produce a ~100 fs laser pulse train at a repetition rate of 1 KHz. The probe beam was provided by white light generated by focusing a small amount of the fundamental beam on a sapphire plate (0.9–2.7 eV) or on a H₂O cell (1.1–3.2 eV). Measurements were performed such that the pump beam irradiated a larger area of the sample (~0.25 mm²) compared to the probe beam (~0.1 mm²) in order to guarantee a homogeneous optical density throughout the probing area. The instrument response of the setup is ~200 fs. During the

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1 experiments the samples were kept at 80 K in a cryostat
 2 flowed with nitrogen and under atmospheric pressure.

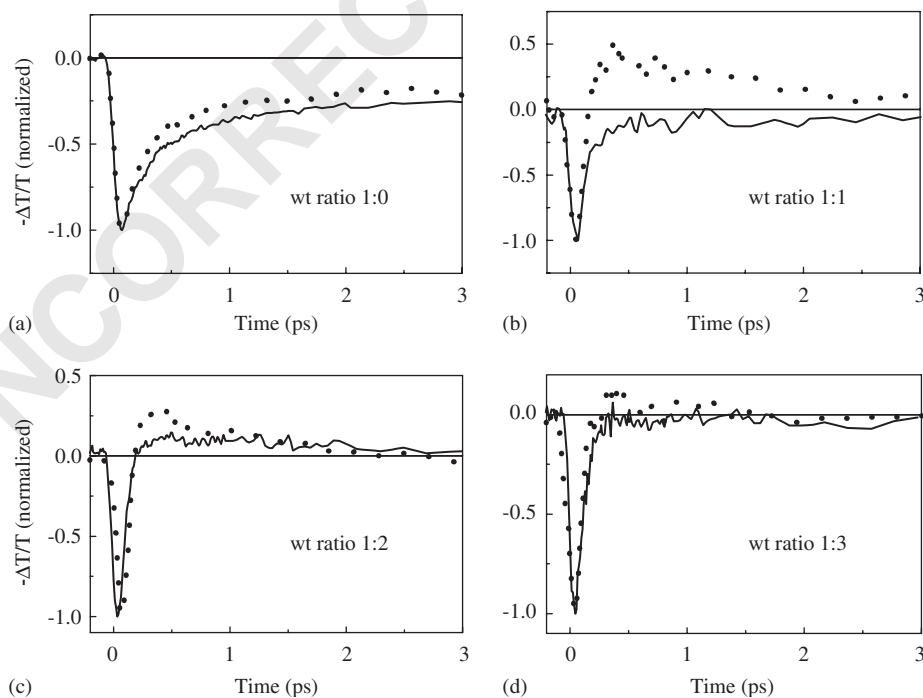
3 3. Results and discussion

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 7 The excitation energy in the femtosecond transient
 8 absorption experiments was kept at 2.2 eV to minimize
 9 the influence of photo-degradation effects on the measure-
 10 ments and excitons are created. Probing at 2.1 eV reveals
 11 the changes in absorption due to either stimulated emission
 12 (SE) of the exciton and/or exciton–exciton annihilation
 13 (E–EA). In the pristine conjugated polymer a negative
 14 signal is observed (wt ratio 1:0, see Fig. 1a) and the
 15 transient comprises an ultrafast decay ~ 500 fs and a long
 16 component with several hundred picoseconds under
 17 relatively low excitation intensity ($\sim 0.8 \mu\text{J}/\text{pulse}$). The
 18 negative signal at 2.06 eV, (600 nm) was argued to be
 19 dominated by SE of the exciton. The ultrafast component
 20 of the SE is characteristic of E–EA and the long
 21 component is typical of the exciton lifetime. Increasing
 22 the excitation intensity ($\sim 2.5 \mu\text{J}/\text{pulse}$) results in the
 23 shortening of the ultrafast component to ~ 280 fs. This is
 24 in line with the E–EA assignment.

25 Fig. 1b illustrates transients at 2.1 eV for
 26 MDMO–PPV:ZnO (wt ratio 1:1). At the low excitation
 27 intensity the transient behaves similar to that of pristine
 28 MDMO–PPV polymer film as shown in Fig. 1a, except that
 29 the relative amplitude of the long component of SE is
 30 smaller and the ultrafast component is even faster to
 31 ~ 100 fs. The weakening of the long component of SE can

32 be explained by the fact that the blend is in fact a mixture
 33 of two phases: one is the pristine MDMO–PPV, which is
 34 dominant in the areas without ZnO and the other is
 35 MDMO–PPV:ZnO blend phase being dominant in the
 36 areas where ZnO is in the vicinity of MDMO–PPV
 37 conjugated polymer chains. In the blend phase electron
 38 transfer occurs at the interfaces between MDMO–PPV
 39 conjugated polymer chains and ZnO nanoparticles. The
 40 excitons in the blend phase also have contributions to the
 41 transient at 2.1 eV and their lifetime will be limited by the
 42 electron transfer process from MDMO–PPV to ZnO,
 43 which is known to be less than 100 fs when the electron
 44 acceptor is PCBM [5]. We assume that the electron transfer
 45 time will be similar when the electron acceptor is ZnO. The
 46 relative amplitude of the long SE component is therefore
 47 smaller compared to pristine MDMO–PPV films (Fig. 1a).
 48 Also the time constant of the ultrafast SE component will
 49 be shorter than that in the pristine film due to the existence
 50 of the excitons of ultra-short lifetime in the blend phase.

51 Increasing the excitation intensity results in a change in
 52 the decay kinetics. As can be seen in Fig. 1b, the signal
 53 changes from negative to positive. Apart from SE and
 54 E–EA, also polaron absorption is present at the probe
 55 energy. This polaron absorption is responsible for the extra
 56 component with time constant of ~ 2 ps (Fig. 1b). Accord-
 57 ing to the discussion above the signal at 2.1 eV is a
 58 superposition of exciton SE and polaron absorption in the
 59 blend phase and exciton SE in pristine phase. In the
 60 pristine phase the transient at 2.1 eV should reflect merely
 61 the feature of pristine polymer similar to Fig. 1a, while in



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 57 Fig. 1. Kinetics of transient absorption at 2.1 eV of MDMO–PPV:ZnO blend films with excitation at 2.2 eV. $T = 80$ K. Excitation intensities are $0.8 \mu\text{J}/\text{pulse}$ (solid curves) and $2.5 \mu\text{J}/\text{pulse}$ (dotted curves), respectively.

1 the blend phase electron transfer occurs and in conse-
2 quence polarons are generated. The fact that a positive
3 component appears at high excitation intensity reveals the
4 excitation intensity dependence of the polaron generation
5 in the blend phase. This observation suggests that besides
6 the exciton decomposition, E-EA is another channel to
7 generate polaron. Therefore, the polaron number can be
8 regarded as being proportional to the number of excitons,
9 and thus has a linear function of the excitation intensity,
10 while saturation effect manifests itself in an underlinear
11 function of the excitation intensity in the exciton absorp-
12 tion. The short lifetime of the polaron (~ 2 ps) implies that
13 in the blend phase the polaron cannot diffuse or transport
14 long before it recombines, e.g. with negatively charged
15 ZnO. More ZnO in the MDMO-PPV:ZnO blend increases
16 the area of blend phase relative to the pristine polymer
17 phase as is reflected in the transients of blend 1:2 shown in
18 Fig. 1c. At relatively low excitation intensity a positive
19 signal already appears with a time constant ~ 2 ps, in
20 contrast to the case in blend 1:1 (Fig. 1b).

21 This might generate the expectation that stronger
22 absorption of the polaron would show up at low excitation
23 intensity if more ZnO were in the blend. This, however,
24 turns out to be not the case. Higher ZnO concentration
25 quenches the spectroscopic signature of the polaron. Fig. 1d
26 shows the transient at 2.1 eV of a MDMO-PPV:ZnO blend
27 with weight ratio 1:3. It can be seen that only the ultrafast
28 SE is apparent. Neither the long SE component nor the
29 polaron absorption is significant. Disappearance of the
30 polaron absorption may be due to several reasons, e.g. (1)
31 lack of the blend phase or (2) ultrafast back electron
32 transfer. Indeed, the scanning electron microscopic picture
33 of the blend 1:3 shows holes in ZnO area, while relatively
34 homogeneous structure is shown in the SEM pictures of the
35 blend films 1:1 and 1:2. This would imply that in blend film
36 1:3 the blend phase area has been reduced compared to the
37 blend films 1:1 and 1:2. On the other hand the ultrafast SE
38 ~ 100 fs, differing from the pristine MDMO film, indicates
39 that polarons are generated in the blend film 1:3. Thus
40 absence of the polaron absorption could also be argued to

41 come from the speed up of the back electron transfer,
42 which is consistent with the observation that the excitons in
43 the blend phase live ultrashortly ~ 100 fs. We infer that
44 both factors exist in blend film 1:3.

45 4. Conclusion 49

50 In summary, the photogeneration of polarons resulting
51 from electron transfer from MDMP-PPV conjugated
52 polymer chains to ZnO at 80 K, strongly depends on the
53 weight ratio between MDMP-PPV and ZnO and on the
54 excitation intensity. It is confirmed from the femtosecond
55 transient absorption kinetics at 2.1 eV that the polarons are
56 generated within the excitation laser pulse (~ 100 fs) and
57 have a lifetime of ~ 2 ps in MDMO-PPV:ZnO blends. Both
58 pristine polymer and blend phase coexist in the blend films
59 1:1 and 1:2. Increasing the excitation intensity favors the
60 generation of the polaron. Quenching of the absorption of
61 the polaron in high ZnO concentration is ascribed to the
62 lack of the blend area and the speed up of back electron
63 transfer.

64 Acknowledgments 67

68 This work is partly supported by the Dutch Polymer
69 Institute project “Nanocomposite hybrid photovoltaic
70 cells”.

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