## Rational Design and Reticulation of Infinite qbe Rod Secondary Building Units into Metal-Organic Frameworks through a Global Desymmetrization Approach for Inverse $C_3H_8/C_3H_6$ Separation

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#### **1. Materials and General Procedures.**

All the chemicals are commercially available and were used without further purification. Thermogravimetric analyses (TGA) were performed on a TGA/DCS 1 system (Mettler-Toledo, Columbus, OH) with STARe software. Samples were heated from 25 to 600 °C at a rate of 5 °C/min under N<sub>2</sub> with flow rate 20 mL/min. Powder X-ray diffraction (PXRD) data were collected on a STOE-STADI P powder diffractometer operating at 40 kV voltage and 40 mA current with Cu-K $\alpha$ 1 X-ray radiation ( $\lambda$  = 0.154056 nm) in transmission geometry. The calculated PXRD patterns were produced using the Mercury software and single crystal reflection data. <sup>1</sup>H NMR, and <sup>13</sup>C NMR experiments were carried out on a Bruker Avance III 500 MHz system. N<sub>2</sub> sorption isotherms were measured using a Micrometritics ASAP 2420 surface area analyzer at 77 K. Before the isotherm measurement, Cu-ASY, UMCM-150, and UMCM-151 were exchanged with methanol 3 times (6 h for each exchange) and acetone 3 times (6 h for each exchange) then degassed on ASAP 2420 for 10 h at 100 °C; SEM images were acquired using a Hitachi SU8030 scanning electron microscope. Prior to imaging, the sample was coated with OsO4 (~9 nm) in a Denton Desk III TSC Sputter Coater. CO<sub>2</sub>, C<sub>3</sub>H<sub>8</sub>, and C<sub>3</sub>H<sub>6</sub> sorption isotherms were measured on a Micromeritics 3Flex.

#### 2. Micro electron diffraction crystallography.

Within the last twenty years, the use of electron diffraction to measure requisite diffraction data and intensities to model the solid-state structures of molecules and materials that crystallize in the sub-micron regime has become a topic of great interest to the chemical and structural biology communities. Now referred to as 3D ED (three-dimensional electron diffraction) or Micro ED (micro-electron diffraction), the stratagems by which these data are now collected closely mirror those utilized in X-ray diffraction, obviating the necessity that the user possess prior transmission electron microscope operational awareness. Until recently the capability of TEMs to collect electron diffraction data via the addition of various hardware supplements and software packages has been the sole avenue whereby this powerful analytical tool has been demonstrated to the scientific community. While this has yielded some promising results, the absence of a dedicated electron diffractometer that established a simple and easy-to-follow experimental workflow was not universally available and made it difficult for this technique to garner more expansive use as an analytical tool.

Modern dedicated electron diffractometers, such as the Rigaku XtaLAB Synergy-ED, have recently become available and offer a

workflow paralleling that of modern X-ray diffractometers (*CrystEngComm.* **2021**, *23*, 8622). The XtaLAB Synergy-ED unifies the processes of sample screening, data collection and processing, and structure solution into a convenient, easy-to-follow process controlled through the well-known software package *CrysAlis<sup>Pro</sup>*. The XtaLAB Synergy-ED was developed through collaborative efforts involving JEOL and Rigaku and relies on the former's expertise in microscopy in conjunction with the latter's experience in X-ray diffraction to provide a dedicated electron diffractometer capable of collecting high-quality data on both small molecule and protein samples (*Symmetry* **2023**, *15*, 1555).

The electron diffraction data were collected on multiple crystalline, nanometer size grains of Cu-ASY (data sets numbered 4463 to 4474) at room temperature using a XtaLAB Synergy-ED electron diffractometer equipped with a HyPix-ED detector optimized for operation in the electron diffraction experimental setup and a 200kV electron source at a wavelength of 0.025 Å. All measurements completed in less than ten minutes, resulting in a total experiment time of forty minutes. For improved data quality, a total of 6 measurements (4463, 4464, 4465, 4469, 4472, and 4474) were merged to build the final reflection file, resulting in an almost complete data set with a redundancy of 5.8. Data collection

strategies for the individual grains and the merging of their respective datasets were completed using CrysAlis<sup>Pro</sup> [CrysAlisPro, Rigaku Oxford Diffraction, version 171.43.44a, 2022]. Data processing was done using CrysAlis<sup>Pro</sup> and included multi-scan absorption corrections applied using the SCALE3 ABSPACK scaling algorithm [SCALE3 ABSPACK–A Rigaku Oxford Diffraction program for Absorption Corrections, Rigaku Oxford Diffraction, 2017]. All structures were solved via intrinsic phasing methods using ShelXT [Sheldrick, G.M. (2015). Acta Cryst. A71, 3-8] and refined with ShelXL [Sheldrick, G.M. (2015). Acta Cryst. A71, 3-8] within the Olex2 graphical user interface [Dolomanov, O.V., Bourhis, L.J., Gildea, R.J, Howard, J.A.K. & Puschmann, H. (2009), J. Appl. Cryst. 42, 339-341]. The final structural refinement included anisotropic temperature factors on all constituent non-hydrogen atoms. Hydrogen atoms were attached via the riding model at calculated positions using suitable HFIX commands.

Table S1 shows the data collection details; Table S2 features the data quality statistics overview for all selected data collections; Table S3 gives the processing results for the merged data. Figure S0 presents the grain snapshots and diffraction images of all six data measurements.

Data set	4463	4464	4465	4469	4472	4474
Exposure Time	1.25 s	2.0 s	2.0 s	2.0 s	2.0 s	1.25 s
Scan Width	0.25°	0.25°	0.25°	0.25°	0.25°	0.25°
Total Frames	480	400	440	300	320	400
Total Time	10 mn 08 s	13 mn 28 s	14 mn 47 s	10 mn 07 s	10 mn 47 s	8 mn 27 s
Detector Distance	660 mm					
Wavelength	0.0251 Å					
Temperature	293 K					

## Table S1. Data collection details for the six selected grains.

## Table S2. Data quality statistics overview for the selected six

Data set	Resolution	Comp. %	Redund.	<f<sup>2/σ(F<sup>2</sup>)&gt;</f<sup>	R <sub>int</sub>
4463	1.00 Å	65.9%	2.0	6.1	12.8%
4464	1.00 Å	51.9%	1.9	7.5	11.1%
4465	1.03 Å	43.6%	2.0	8.5	12.9%
4469	1.00 Å	38.9%	1.9	3.2	13.0%
4472	1.00 Å	45.0%	2.0	3.2	13.4%
4474	1.00 Å	55.1%	2.0	4.4	18.1%

#### data sets

#### Table S3. Processing results for the merged data.

Resolution	kept	Theory	Unique	Compl. %	Redund.	<f<sup>2/σ(F<sup>2</sup>)&gt;</f<sup>	R <sub>int</sub>	R <sub>pim</sub>	σΒ	CC½
16.44-2.17	1462	289	275	95.2	5.3	34.40	0.091	0.044	0.025	0.998
2.17-1.71	1543	289	282	97.6	5.5	15.98	0.153	0.071	0.062	0.992
1.71-1.50	1678	289	282	97.6	6.0	8.13	0.262	0.115	0.135	0.967
1.50-1.35	1546	289	280	96.9	5.5	6.00	0.303	0.140	0.175	0.961
1.35-1.26	1779	289	283	97.9	6.3	5.17	0.365	0.157	0.186	0.966
1.26-1.19	1665	289	283	97.9	5.9	5.06	0.365	0.163	0.222	0.957
1.19-1.12	1622	289	281	97.2	5.8	3.80	0.469	0.212	0.316	0.854
1.12-1.08	1690	289	282	97.6	6.0	2.62	0.536	0.236	0.413	0.875
1.08-1.04	1700	289	284	98.3	6.0	1.74	0.643	0.283	0.527	0.758
1.04-1.00	1442	297	270	90.9	5.3	1.82	0.617	0.293	0.529	0.770
16.44-1.00	16127	2898	2802	96.7	5.8	8.16	0.230	0.104	0.106	0.995



Figure S0. Grain snapshots and diffraction images of the six data sets.

Crystal data and details of the data collection are given in **Tables S4**. CCDC 2269110 contains the supplementary crystallographic data for this paper. The data can be obtained free of charge from The Cambridge Crystallographic Data Centre via <u>www.ccdc.cam.ac.uk/data\_request/cif</u>. The relatively high  $R_1$  and *wR*<sup>2</sup> values and ADP alerts are due to the poor crystal data collected by ED technique at room temperature, and these poor parameters are commonly observed in other reported crystal data sets (*Nat. Commun.* **2022**, *13*, 415 / *CrystEngComm.* **2021**, *23*, 8622).

Identification Code	Cu-ASY
Empirical formula	C <sub>27</sub> H <sub>16</sub> Cu <sub>2</sub> O <sub>7</sub>
Formula weight	579.5
Temperature (K)	293(2) K
Crystal system	Triclinic
Space group	P-1
Unit cell	$a = 5.528(2)$ Å $\alpha = 77.15(3)$
dimensions	$b = 15.184(6) \text{ Å } \beta = 83.49(4)$
	<i>c</i> = 16.951(7) Å γ = 86.91(4)
Volume (ų), Z	1377.7(10), 2
Density	1.395
<i>F</i> (000)	199
Reflections	16127 / 2802
Completeness to	0.719 / 96.7 %
Rint	0.2515

 Table S4. Crystal data and structure refinements

Refinement	Full-matrix least-squares on F^2
Data / restraints /	2802 / 854 / 218
Goodness-of-fit on	1.247
Final <i>R</i> indices	$R_1 = 0.2869, wR_2 = 0.6678$
R indices (all data)	$R_1 = 0.3683, wR_2 = 0.7456$
Largest diff. peak	0.374 / -0.356

#### 3. Fitting of pure component isotherms

The experimentally collected isotherms for  $C_3H_8$  and  $C_3H_6$  at 273 K, 298, and 313 K in Cu-ASY were fitted with the dual-site Langmuir-Freundlich equation.

$$q = q_{A,sat} \frac{b_A p^{c_A}}{1 + b_A p^{c_A}} + q_{B,sat} \frac{b_B p^{c_B}}{1 + b_B p^{c_B}}$$

The Langmuir parameters for each site is temperature-dependent,

$$b_A = b_{A\ 0} exp\left(\frac{E_A}{RT}\right); b_B = b_{B\ 0} exp\left(\frac{E_B}{RT}\right)$$

where *p* (unit: kPa) is the pressure of the bulk gas at equilibrium with the adsorbed phase, *q* (unit: mmol g<sup>-1</sup>) is the adsorbed amount per mass of adsorbent,  $q_{sat}$  (unit: mmol/g) is the saturation capacities, *c* (unit: kPa<sup>-1</sup>) is the affinity coefficient.

#### Calculation of isosteric heat of adsorption $(Q_{st})$

The Clausius-Clapeyron equation was employed to calculate the heat of adsorption:

$$Q_{st} = RT^2 \left(\frac{\partial \ln p}{\partial T}\right)_q$$

was determined using the pure component isotherm fits using the Virial equation.

#### IAST calculations of adsorption selectivity

For the separation of a binary mixture of components  $C_3H_8$  and  $C_3H_6$ , the adsorption selectivity is defined by

$$S_{ads} = \frac{q_1/q_2}{y_1/y_2}$$

In equation (4), the  $q_1$ , and  $q_2$  represent the molar loadings of C<sub>3</sub>H<sub>8</sub> and C<sub>3</sub>H<sub>6</sub>, expressed in mol kg<sup>-1</sup>, within the MOF that is in equilibrium with a bulk fluid mixture with mole fractions  $y_1$ , and  $y_2 =$ 1- $y_1$ . The molar loadings, also called gravimetric uptake capacities, are usually expressed with the units mol kg<sup>-1</sup>. The IAST calculations of 50/50 mixture adsorption taking the mole fractions  $y_1 = 0.5$  and  $y_2$ = 1- $y_1$ = 0.5 for a range of pressures up to 101 kPa and 298 K were performed.

#### **Breakthrough experiments**

The breakthrough experiments were carried out in a dynamic gas breakthrough setup. A stainless-steel column with inner dimensions of 4 mm and a length of 80 mm was used for sample packing. Microcrystalline samples (0.42 g of Cu-ASY microcrystals) were air dried and then packed into the columns. The columns were activated in vacuum oven at 80°C. The mixed gas flow of binary gas ( $C_3H_6/C_3H_8$  at 50/50, v/v) and pressure were controlled by using a pressure controller valve and a mass flow controller. Outlet effluent from the column was continuously monitored using gas chromatography (GC-2014, Shimadzu) with a thermal conductivity detector. The column packed with activated sample was first purged with helium gas flow for 1 h at room temperature. The mixed gas flow rate during the breakthrough process is 2 mL min<sup>-1</sup> at 1 bar. After the breakthrough experiment, the sample was purged with helium gas flow (40 mL min<sup>-1</sup>) at 298 K, from which the outlet effluent was monitored. The column can be regenerated and purged with helium gas flow (50 mL min<sup>-1</sup>) at 298 K for 20 min before next cycle.

Since Cu-ASY is a C<sub>3</sub>H<sub>8</sub> selective material, the productivity of C<sub>3</sub>H<sub>6</sub> can be directed calculated based on the breakthrough time obtained from the breakthrough experiment. The breakthrough time of C<sub>3</sub>H<sub>6</sub> is 2.2 min/g, the gas flow of C<sub>3</sub>H<sub>6</sub> is 1.0 cm<sup>3</sup>/min, therefore the C<sub>3</sub>H<sub>6</sub> productivity is calculated to be 2.2 min/g \* 1 cm<sup>3</sup>/min = 2.2 cm<sup>3</sup>/g or 2.2 L/kg.



#### 4. Calculation details.

All the calculations were performed in the Material Studio 2019 package (BIOVIA, Dassault Systems, Materials Studio 2019, Dassault Systems, San Diego, 2018). The crystal structure of Cu-ASY was taken as initial geometry for further computational calculations. The partial charge of framework atoms was taken from Mulliken charge and considered as rigid in the simulations. The charges of gas molecules were derived from ESP charge. The binding sites of propylene and propane were first investigated by GCMC-simulated annealing calculations performed for one molecule within a unit cell, which started from an initial temperature of  $1 \times 10^5$  K, followed by  $1.0 \times 10^6$  Monte Carlo steps. Universal forcefield (UFF) was used, in which the interaction energy between gas molecules and frameworks were computed through the Coulomb and Lennard-Jones 6-12 (LJ) potentials. The cut-off radius was chosen as 12 Å for the LJ potential and the long-range electrostatic interactions were handled by the Ewald summation method, with a Buffer width of 0.5 Å and accuracy of  $1 \times 10^{-5}$  kcal mol<sup>-1</sup>. The equilibration steps and the production steps were set as  $1 \times 10^{6}$  and  $1 \times 10^{7}$ , respectively. The preliminary host-guest structure was further allowed to fully relaxed and optimized using first-principle density functional theory (DFT) in the CASTEP code

(Kristallogr., 2005, 220, 567.) The generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) functional (Phys. Rev. Lett. 1996, 77, 3865) and on-the-fly generated ultrasoft pseudopotentials<sup>3</sup> were used. Grimme (G06) semi-empirical methods to describe the long-range van der Waals interactions. A cutoff energy of 650 eV and a 2 × 2 × 2 *k*-point mesh was found to be enough for the total energy to converge within  $1 \times 10^{-5}$  eV atom <sup>-1</sup>. Full geometry optimizations were performed on the structures loaded with one C<sub>3</sub>H<sub>6</sub> or C<sub>3</sub>H<sub>8</sub> molecule. The static binding energy was calculated:  $\Delta E = E(MOF) + E(gas) - E(MOF + gas)$ .

#### 5. Synthesis of organic linkers.

The two  $C_{2v}$  symmetric tritopic linkers were synthesized according to the reported procedures (*J. Am. Chem. Soc.* 2007, *129*, 15740 / *J. Am. Chem. Soc.* 2010, *132*, 13941). The  $C_s$ symmetric tritopic linker (H<sub>3</sub>L) was synthesized according to the reported procedures (*Chem. Eur. J.* 2015, *21*, 5954) and the synthetic scheme was shown below.









#### 6. Synthesis of Cu-MOFs.

**UMCM-150**: 100 mg Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O and 50 mg biphenyl-3,4',5-tricarboxylate were added into a mixture of DMF/dioxane/H<sub>2</sub>O (4 mL/1 mL/1 mL) in a 2-dram vial. The solution was sonicated for 5 minutes and then placed into an 80  $^{\circ}$ C oven for 2 days. Large blue hexagonal crystals were collected for further characterizations and measurements.

**UMCM-151**: 100 mg Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O and 50 mg 3,5-di(4'-carboxylphenyl)benozoic acid were added into a mixture of DMF/dioxane/H<sub>2</sub>O (4 mL/1 mL/1 mL) in a 2-dram vial. The solution was sonicated for 5 minutes and then placed into an 80  $^{\circ}$ C oven for 2 days. Large blue block crystals were collected for further characterizations and measurements.

**Cu-ASY:** 100 mg Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O and 50 mg H<sub>3</sub>L were added into a mixture of DMF/dioxane/H<sub>2</sub>O (4 mL/1 mL/1 mL) in a 2-dram vial. The solution was sonicated for 5 minutes and then placed into an 80  $^{\circ}$ C oven for 2 days. Blue powders were collected for further characterizations and measurements.

#### 7. Figures S1-S3. Additional X-ray crystallographic structures

There is a strategy to assess the unicity of a net. Following the rules described in the CG&D paper "Deconstruction of Crystalline Networks into Underlying Nets: Relevance for Terminology Guidelines and Crystallographic Databases" Cryst. Growth Des., **2018**, *18*, 3411. It is possible to analyze any structure with at least 2 representations: (1) Standard = the nodes are the single point metals and the other nodes are what remain i.e. single atoms like oxygens or multi branched ligands. This representation is always possible and is very useful for data base retrieval of isoreticular compounds; (2) Cluster all or single node: this representation is tricky for rod-MOF because the cluster algorithm will find an infinite SBU, so the researchers need to decide how to simplify/describe such infinite SBU. The modes are with the Point-of-Extension (PE) introduced by Mike O'Keeffe in ref 15, and as PE&M or STR as in ref 45 in the main text.

The ToposPro team developed a database of standard description of MOF as they are reported in CSD, and updated every year. The database is Topcryst: <u>https://topcryst.com/</u> (see the open access ref: <u>https://doi.org/10.1080/27660400.2022.2088041</u>). So, we can search our MOF by the standard underlying nets, and we found only two structures (refcode: FAHNOT and KUZBAL) that are

chemically very different from ours, not rod-MOF but containing SBU dimeric  $Cu_2$  and  $Zn_2$ . So, we are confident that our rod-MOF is unique.

About the 4<sup>4</sup>(3,3) *qbe* tubule, it has been observed only in three metal-organic nanotubes (*Inorg. Chem. Commun.* 2008, *11*, 616 with refcode of AFOGAE), (*Inorg. Chem.* 2015, *54*, 6680 with refcode of BUGXIM), and (*Angew. Chem., Int. Ed.* 2009, *48*, 9461 with refcodes of SUKNOC and SUKPAQ), and never been reported in MOFs to date.



**Figure S1**. The rod SBU with Cu square pyramidal in blue, Cu octahedral in green and the O of the OH group in yellow. There are three distinct carboxylate, two  $\mu^2$  and one  $\mu^3$ . Bottom some views of the tube SBU **qbe** / 4<sup>4</sup>(3,3) in blue obtained following the PE (point-of-extension) rules that join the C atoms of the carboxylate from the ligands.



**Figure S2.** Views of the underlying nets together with the crystal structure, left: the SBU **qbe** /  $4^4(3,3)$  and right: the 3,5-c **gfc** net.



**Figure S3.** (Top) two alternative description of the underlying net from the crystal structure: left the 3,5-c **gfc** (PE description) right 3,5-c **gra** (STR description). (Bottom) The ideal **gra** net showing the ABAB stacking of **hcb** graphene like layers

#### 8. Figures S4-S6. Additional PXRD patterns



**Figure S4.** PXRD patterns of Cu-ASY synthesized in different batches, showing the consistence of phase purity.



**Figure S5.** PXRD patterns of UMCM-150, showing its bulk phase purity. The difference of peak intensities at about 4, 7, and 13.5 degrees are the result of preferred orientation effect coming from the very thin crystal morphology.



**Figure S6.** PXRD patterns of UMCM-151, showing its bulk phase purity and partial structural collapse upon activation.

### 9. Figure S7-S10. Additional CO<sub>2</sub> and N<sub>2</sub> sorption isotherms



**Figure S7.** N<sub>2</sub> sorption isotherms of Cu-ASY synthesized in different batches.



**Figure S8.** CO<sub>2</sub> sorption isotherms of Cu-ASY, UMCM-150, and UMCM-151 at room temperature and ambient pressure.



Figure S9. N<sub>2</sub> sorption isotherms of UMCM-150.



Figure S10. N<sub>2</sub> sorption isotherms of UMCM-151.

## 10. Figure S11-24. Additional C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> sorption isotherms



**Figure S11.**  $C_3H_6$  and  $C_3H_8$  adsorption/desorption isotherms of Cu-ASY at different temperatures.



**Figure S12.**  $C_3H_8$  adsorption isotherms of Cu-ASY synthesized in different batches.



**Figure S13.**  $C_3H_6$  adsorption isotherms of Cu-ASY synthesized in different batches.



**Figure S14.** IAST selectivity calculations of Cu-ASY samples synthesized in different batches.



**Figure S15.**  $C_3H_6$  and  $C_3H_8$  sorption isotherms collected at 298 K and 1 bar of UMCM-150.



**Figure S16.** Enlarged  $C_3H_6$  and  $C_3H_8$  sorption isotherms of UMCM-150.



**Figure S17.** C<sub>3</sub>H<sub>8</sub> sorption isotherms of UMCM-150 at different temperatures.



**Figure S18.**  $C_3H_6$  sorption isotherms of UMCM-150 at different temperatures.



Figure S19. IAST selectivity of UMCM-150 for equivalent  $C_3H_6/C_3H_8$  gas mixture.



**Figure S20.**  $C_3H_6$  and  $C_3H_8$  sorption isotherms collected at 298 K and 1 bar of UMCM-151.



**Figure S21.** Enlarged  $C_3H_6$  and  $C_3H_8$  sorption isotherms of UMCM-151.



**Figure S22.** C<sub>3</sub>H<sub>8</sub> sorption isotherms of UMCM-151 at different temperatures.



**Figure S23.**  $C_3H_6$  sorption isotherms of UMCM-151 at different temperatures.



Figure S24. IAST selectivity of UMCM-151 for equivalent  $C_3H_6/C_3H_8$  gas mixture.

## 11. Figure S25. TGA curves



## 12. Figures S26-S28. Optical and SEM images



Figure S26. SEM images of Cu-ASY microcrystals



Figure S27. Optical (Top) and SEM images (bottom) of UMCM-150.



Figure S28 Optical (Top) and SEM images (bottom) of UMCM-151.

## 13. Figure S29. Comparison of C<sub>3</sub>H<sub>8</sub>/C<sub>3</sub>H<sub>6</sub> separation selectivity



**Figure S29.** Comparison of inverse  $C_3H_8/C_3H_6$  selectivity of Cu-ASY with other  $C_3H_8$ -selective materials.

## 14. Figure S30. SEM and EDS mapping images of Zn-ASY



C Kα1\_2

## 15. Figure S31. PXRD patterns of Zn-ASY and Cu-ASY



## 16. Figure S32. Derived 2D fingerprint plot of host-guest

## interactions in Cu-ASY



#### 17. Figure S33. Comparison of C<sub>3</sub>H<sub>6</sub> productivity



18. Figure S34. The PXRD patters of Cu-ASY after different treatments



19. Figure S35. The sorption isotherms of  $C_2H_4$  and  $C_2H_6$  for Cu-ASY at 298 K



20. Figure S36. Water sorption isotherm of Cu-ASY at 298 K



21. Figure S37. The breakthrough experiment of Cu-ASY under

30% RH



# 22. Figure S38. Structureless refinement (Le Bail) of the powder XRD data.



#### 23. Figure S39. The fitting results of isotherms









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