

## Article

# Tungsten Trioxide and Its TiO<sub>2</sub> Mixed Composites for the Photocatalytic Degradation of NO<sub>x</sub> and Bacteria (*Escherichia coli*) Inactivation

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**Abstract:** The increased air pollution and its impact on the environment and human health in several countries have caused global concerns. Nitrogen oxides (NO<sub>2</sub> and NO) are principally emitted from industrial activities that strongly contribute to poor air quality. Among bacteria emanated from the fecal droppings of livestock, wildlife, and humans, *Escherichia coli* is the most abundant, and is often associated with the health risk of water. TiO<sub>2</sub>/WO<sub>3</sub> heterostructures represent emerging systems for photocatalytic environmental remediation. However, the results reported in the literature are conflicting, depending on several parameters. In this work, WO<sub>3</sub> and a series of TiO<sub>2</sub>/WO<sub>3</sub> composites were properly synthesized by an easy and fast method, abundantly characterized by several techniques, and used for NO<sub>x</sub> degradation and *E. coli* inactivation under visible light irradiation. We demonstrated that the photoactivity of TiO<sub>2</sub>/WO<sub>3</sub> composites towards NO<sub>2</sub> degradation under visible light is strongly related to the WO<sub>3</sub> content. The best performance was obtained by a WO<sub>3</sub> load of 20% that guarantees limited e<sup>-</sup>/h<sup>+</sup> recombination. On the contrary, we showed that *E. coli* could not be degraded under visible irradiation of the TiO<sub>2</sub>/WO<sub>3</sub> composites.

**Keywords:** photocatalysts; visible light; tungsten trioxide; composites

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## 1. Introduction

Due to growing industrialization, urban environments have faced chronic air pollution issues in the last decades. Exhaust gases and burning fuels from factories represent the primary sources of air pollutants on a global scale, causing a significant impact on human health, animal and plant life, and climate [1]. Although natural sources responsible for air pollutants production, such as broad forest fires, volcanic eruptions, and soil erosion, can play a role in air pollution, the emissions resulting from human activities, such as motor vehicle exhaust, combustion of fossil fuels, and industrial processes, are the most active and concerning cause of air quality decline [2].

Nitrogen monoxide (NO) and nitrogen dioxide (NO<sub>2</sub>), known as nitrogen oxides (NO<sub>x</sub>), are relevant pollutants whose emissions are directly related to human health problems [3], as they affect respiratory and immune systems [4], to the production of tropospheric ozone, acid rains, and in general to global air pollution.

Over the years, different techniques have been developed for NO<sub>x</sub> abatement. Among the traditional techniques, selective catalytic reduction (SCR) with ammonia in

the presence of oxygen is the most used, mainly applied to reduce  $\text{NO}_x$  emission from combustion processes [5], as well as absorption, adsorption, or electrical discharge processes [6]. However, all these methods are characterized by several limitations and disadvantages that make actual application hard. Moreover, the growing environmental constraints invoke restrictions regarding  $\text{NO}_x$  emission, requiring more efficient techniques for  $\text{NO}_x$  abatement.

In addition, awareness about the importance of supplying adequate drinking water has recently increased. In 2012, the United Nations estimated that nearly 11% of the world's population did not have access to improved drinking water sources. African water resources indeed contain high levels of microbial pathogens, including bacteria, viruses, and protozoa, as well as chemical contaminants. *Escherichia coli* and related bacteria constitute approximately 0.1% of gut flora, and fecal–oral transmission is the primary route through which pathogenic strains of the species cause disease. For that reason, new disinfection technologies are currently in development to fulfill the WHO Guidelines for drinking-water quality (World Health Organization, 2008). The traditional disinfection methods lead to chloro-organic disinfection by-products (DBPs) with carcinogenic and mutagenic effects.

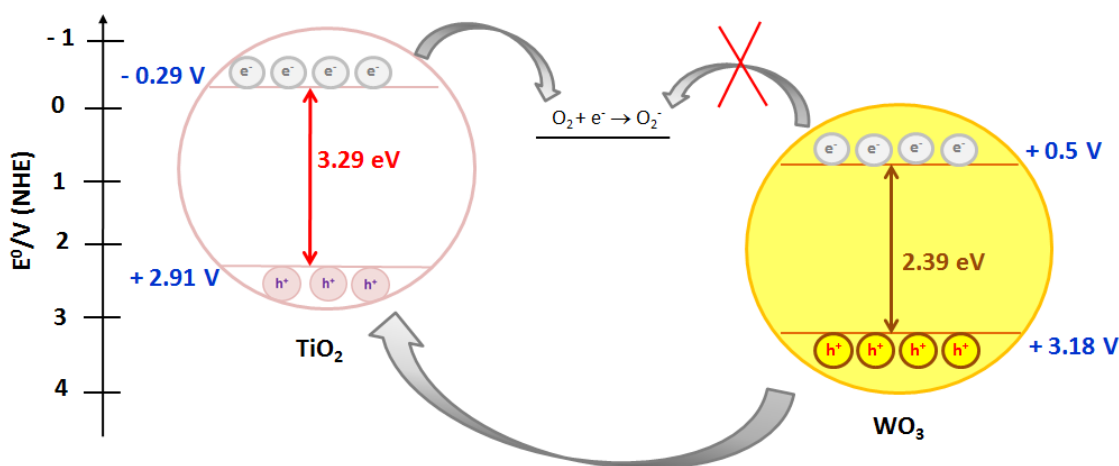
In both study cases, using a  $\text{TiO}_2$  semiconductor as a catalyst under UV or visible irradiation seems the most promising method.

Titania ( $\text{TiO}_2$ ) has been considered the most efficient photocatalyst for a wide variety of applications, such as pollution abatement [7,8], water and air purification [9], antimicrobial applications [10], and energy conversion [11]. However,  $\text{TiO}_2$  in its photoactive anatase phase has a wide band gap of 3.2 eV, limiting the photoactivity of the semiconductor only under UV irradiation [12]. Moreover, because of its suspected carcinogenic nature [13], researchers are willing to replace  $\text{TiO}_2$  with new low-cost and visible-light-active smart materials.

Though many studies have focused on using  $\text{TiO}_2$  [14–18],  $\text{WO}_3$  and its composites have been poorly investigated to date [19–22].

$\text{WO}_3$  is a cheap, physiochemically stable, and mechanically robust semiconductor with a narrow band gap energy (2.4–2.8 eV), making it a visible-light-responsive photocatalyst for different applications [23–30]. Therefore,  $\text{WO}_3$  represents a suitable choice for photocatalytic degradation under visible light irradiation.

As described in Figure 1, because the VB (valence band) edge potential of  $\text{WO}_3$  is lower than that of  $\text{TiO}_2$ , upon photon absorption, electrons can be transferred from the conduction band of  $\text{TiO}_2$  to  $\text{WO}_3$ , whereas photogenerated holes move in the opposite direction from electrons.



**Figure 1.** Photocatalytic mechanism of  $\text{WO}_3$ -loaded  $\text{TiO}_2$  under light irradiation.

The transfer of photogenerated carriers is accompanied by consecutive  $W^{6+}$  reduction into  $W^{5+}$  by capturing photogenerated electrons at trapping sites in  $WO_3$ . In addition,  $W^{5+}$  ions on the surface of  $WO_3$  are reoxidized into  $W^{6+}$ . However, the reduction potential value for the photogenerated electrons in the conduction band is not high enough for the single electron reduction in  $O_2$  (Figure 1) [30]. The holes accumulated in the  $TiO_2$  VB take part in the oxidation process to make  $OH^-$  or  $OH\cdot$  hydroxyl radical reactive species. These processes in  $WO_3/TiO_2$  heterostructures restrain the recombination of electron–hole pairs significantly.

Different approaches have been developed to produce highly active  $TiO_2/WO_3$  composites aiming to optimize the  $WO_3$  content [31]. It has been demonstrated that the influence of  $WO_3$  on  $TiO_2$  photoactivity depends on several factors, such as crystal phase, electrons accumulation ability of  $WO_3$ , type of pollutants, and degradation pathways involved [31].

Yang et al. investigated the role of amorphous  $WO_x$  species, demonstrating that they are more active than the crystalline ones toward methylene blue degradation [32].

Other experiments by Žerjav et al. explained the correlation between the photocatalytic performance of  $TiO_2/WO_3$  and their shallow and deep electron trapping states [33].

However, concerning  $WO_3$  and its mixed oxides, the obtained results are conflicting because in the same cases, the presence of  $WO_3$  seems to positively affect the photoactivity and the performances of  $TiO_2$ . In other cases, the results worsen [19,31,32,34].

Regarding  $NO_x$  degradation, Luévano-Hipólito et al. demonstrated that  $WO_3$  with a polyhedral shape leads to 50%  $NO$  oxidation to  $NO_2$  [19]. On the other hand, Yu and coworkers noticed for the first time the photo-transformation of  $NO_2$  into  $NO$  in the presence of  $N_2$  on the surface of a  $WO_3$  photocatalyst under UV/visible light irradiation [20].

Recently, Mendoza et al. proposed  $TiO_2/WO_3$  composites as efficient materials for  $NO_x$  abatement under visible light, leading to 90% of photodegradation in 1 h [22], whereas Paula and coworkers observed the decay of the photocatalytic activity of  $TiO_2/WO_3$  heterostructures as a function of the  $W(VI)$  content [31].

Jawwad A. Darr et al. reported the easy disinfection of water by  $TiO_2/WO_3$  mixed composites, which induce bacterial inactivation after 30 min of photo-irradiation [35].

In order to clarify the behavior of  $WO_3$  and  $TiO_2/WO_3$  heterostructures in the photocatalytic degradation of  $NO_x$  under visible light irradiation, in this work,  $WO_3$  and a series of  $TiO_2/WO_3$  composites were synthesized by a fast and cost-effective chemical procedure and tested for the photodegradation of  $NO_x$  and the inactivation of *E. coli* under visible light irradiation.

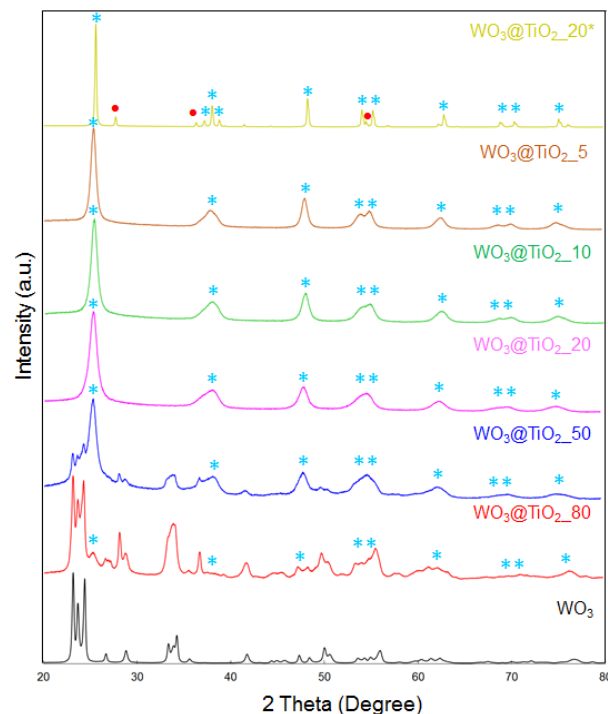
The role of the calcination temperature in the  $TiO_2/WO_3$  preparation has been investigated and critically discussed, as well as the effect of the  $WO_3$  loading in the final composites. Differently from the recent literature, the results proved that high calcination temperatures could cause complete or partial  $WO_3$  sublimation with adverse effects on the activity of the  $TiO_2/WO_3$  heterostructures.

Finally, while the synthesized catalysts were active in  $NO_2$  photodegradation, they were inert to the antibacterial activity under visible light irradiation, in line with the scientific literature [36].

## 2. Results

### 2.1. Materials Characterization

Figure 2 shows the XRD patterns of all the synthesized materials.



**Figure 2.** XRD patterns of the samples of Table1 (\* peaks of TiO<sub>2</sub> anatase, • peaks of TiO<sub>2</sub> rutile).

TiO<sub>2</sub> exhibits the characteristics of diffraction peaks of anatase, as confirmed by the peaks at 25.3°, 37.7°, 48.0°, 53.8°, and 55.0°, with (101), (004), (200), (105), and (211) diffraction planes, respectively. The XRD pattern of WO<sub>3</sub> shows a crystalline phase characterized by diffraction peaks at 23.1°, 23.6°, 24.4°, and 34.2°, corresponding to the (002), (020), (200), and (202) crystal planes of monoclinic phase.

As expected, in the WO<sub>3</sub>@TiO<sub>2</sub> composites, the intensity of the diffraction peaks of WO<sub>3</sub> declines by decreasing the percentage of WO<sub>3</sub>; on the contrary, anatase peaks intensity increases or appears with the higher concentration of TiO<sub>2</sub> in each composite.

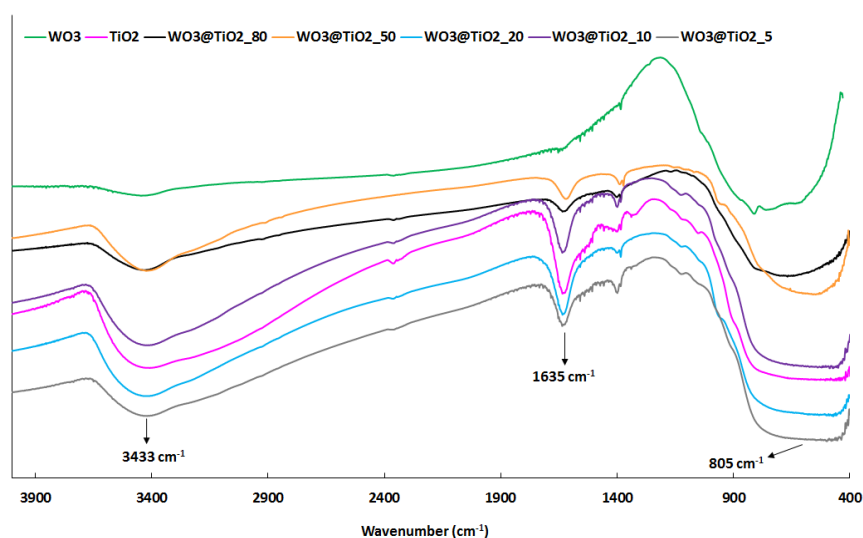
Based on its XRD pattern, WO<sub>3</sub>@TiO<sub>2</sub>\_20\* exhibits higher crystallinity degree if compared to the others and the appearance of new diffraction peaks can be observed.

As reported in the literature [37], it is directly correlated to the high temperature (600 °C) used for the calcination of this material. The degree of crystallinity increases with the temperature, and in the case of TiO<sub>2</sub>-based compounds, at 600 °C the phase conversion from anatase to rutile starts.

According to the literature [22,34], when the WO<sub>3</sub> content in the composite materials is 20% or lower, the diffraction peaks of this semiconductor are undetected. Some authors justify this result with the presence of highly dispersed WO<sub>3</sub> small particles in TiO<sub>2</sub>/WO<sub>3</sub> composites, which makes it hard to detect them by this technique [22,34]. In addition, Yang et al. demonstrated that when the loading amount of WO<sub>3</sub> was below 3 mol%, it exists in highly dispersed amorphous species that do not respond to XRPD. However, accurate quantification of WO<sub>3</sub> loading on TiO<sub>2</sub> after calcination is necessary to verify unequivocally the WO<sub>3</sub>/TiO<sub>2</sub> composite formation rather than a superficial W doping on the TiO<sub>2</sub> surface. In the present work, this was easily carried out by the reaction yield calculation (Equation (2)) and by EDS analysis (Table S1) for two composites with a nominal WO<sub>3</sub> load of 20% (WO<sub>3</sub>@TiO<sub>2</sub>\_20 and WO<sub>3</sub>@TiO<sub>2</sub>\_20\*) calcined at two different temperatures (400 °C and 600 °C). From the results of the reaction yield, the

WO<sub>3</sub>@TiO<sub>2</sub>\_20\* composite calcined at 600 °C exhibits a mass loss of approximately 40%, unlike the same sample calcined at 400 °C, obtained with a yield of 94.8%. Since WO<sub>3</sub> is a low-temperature sublimation material [38], it cannot be excluded that by increasing the calcination temperature, a complete or partial WO<sub>3</sub> sublimation can occur, as also confirmed by the EDS results (Table S1), where the measured percentage of WO<sub>3</sub> in the final composite is lower than 3%, whereas the material calcined at 400 °C shows a 27% of WO<sub>3</sub>.

Figure 3 displays the FT-IR for all the synthesized composites calcined at 400 °C.



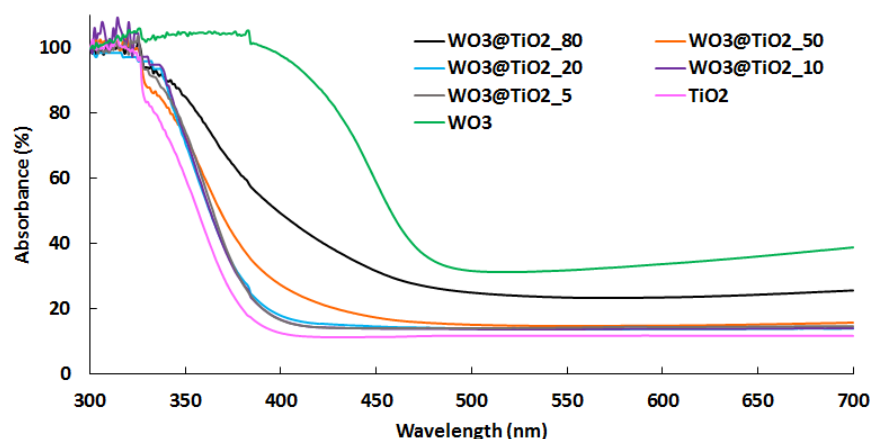
**Figure 3.** FT-IR spectra of the samples calcined at 400 °C.

The FT-IR spectrum of WO<sub>3</sub> exhibits characteristic vibration bands, such as those at approximately 3433 cm<sup>-1</sup> and 1635 cm<sup>-1</sup>, that can be associated with the symmetric stretching vibrations of WO<sub>3</sub> and intercalated water molecules and the deformation vibrations of H–O–H bonds of the adsorbed water molecules, respectively, and the signal at 805 cm<sup>-1</sup>, attributed to O–W–O stretching modes of WO<sub>3</sub> [39].

On the other hand, the FT-IR spectrum of TiO<sub>2</sub> nanoparticles is characterized by several peaks. The OH stretching mode of the hydroxyl groups is responsible for the broad band observed in the range of 3600–3000 cm<sup>-1</sup>, indicating the presence of moisture in the sample. The band at approximately 1605 cm<sup>-1</sup> is due to the OH bending vibrations of the adsorbed water molecules. Finally, the broadband between 1000 and 500 cm<sup>-1</sup> can be related to the Ti–O stretching and Ti–O–Ti bridging stretching modes [40].

As expected, in the FT-IR spectra of the composites, the WO<sub>3</sub> characteristic bands are covered by the more intense ones of TiO<sub>2</sub>.

The optical properties of the synthesized WO<sub>3</sub>/TiO<sub>2</sub> series calcined at 400 °C, as well as of single-phase semiconductors, were investigated by UV–Vis scanning spectrophotometry (Figure 4).



**Figure 4.** UV-Vis absorption spectra of the  $\text{WO}_3/\text{TiO}_2$  composites series calcined at 400 °C.

The main absorption edges of the samples are all around 400 nm, attributing to the excitation of electrons from the valence band to the conduction band. As reported in the literature [41], the empty orbitals of  $\text{W}^{6+}$  (W 5d) are closed to the Ti 3d orbitals of the conduction band. Therefore, the  $\text{O}_2 \rightarrow \text{W}^{6+}$  charge transfer transitions are overlapped with the  $\text{O}_2 \rightarrow \text{Ti}^{4+}$  charge transfer transitions. Increasing the  $\text{WO}_3$  loading, the absorption edge of photocatalysts red-shifts. If this is only slightly noticeable up to 10%, increasing the  $\text{WO}_3$  percentage to 50% and above, the effect is much more evident, and the absorption edges of these materials are shifted at higher wavelengths.

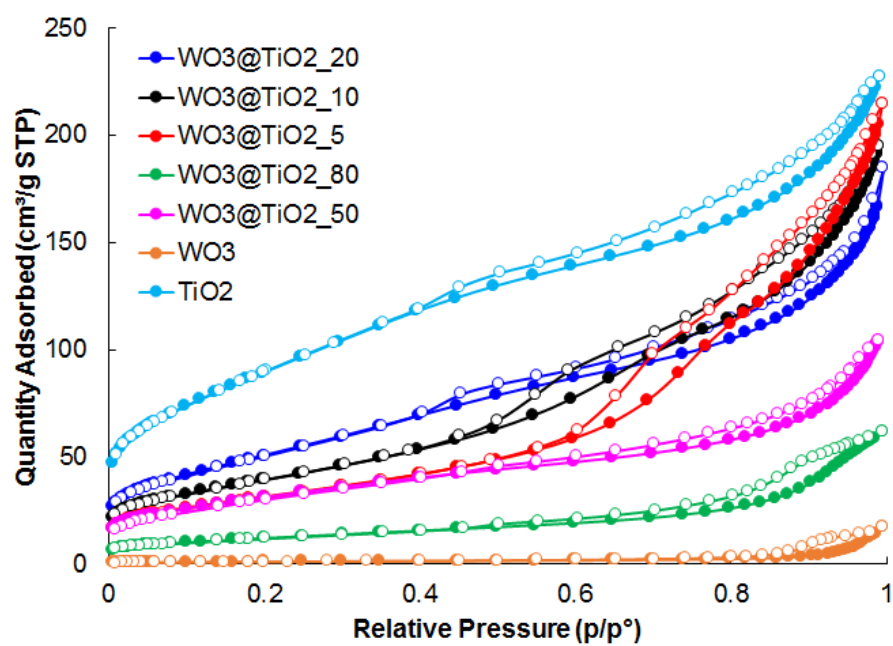
The band gaps of the materials, estimated by the Kubelka–Munk function, are summarized in Table 1, and the Tauc plots are reported in Figure S2.

**Table 1.** Energy of band gap (eV), specific surface area, CBET,  $V_m$ , and mean pore diameter of the  $\text{WO}_3/\text{TiO}_2$  composites series calcined at 400 °C. \* Surface area by BET equation (2-parameters), \*\* mean pore diameter by BJH model from isotherm desorption branch ( $0.3 < p/p_0 < 0.95$ ).

Sample	Band Gap (eV)	* Specific Surface Area ( $\text{m}^2/\text{g}$ )	CBET	$V_m$ ( $\text{cm}^3/\text{g}$ )	** Mean Pore Diameter (nm)
$\text{WO}_3$	2.39	4.00	75.75	0.94	21.17
$\text{WO}_3/\text{TiO}_2_{80}$	2.63	42.78	123.22	9.92	8.6
$\text{WO}_3/\text{TiO}_2_{50}$	3.05	110.65	98.11	25.87	6.4
$\text{WO}_3/\text{TiO}_2_{20}$	3.14	179.78	75.50	43.26	6.0
$\text{WO}_3/\text{TiO}_2_{10}$	3.26	139.47	94.7	33.15	6.8
$\text{WO}_3/\text{TiO}_2_5$	3.20	111.08	112.87	31.29	9.1
$\text{TiO}_2$	3.29	318.00	84.48	75.50	4.70

Samples  $E_g$  decreases, increasing the tungsten content, due to the formation of defective energy levels within the forbidden band gap of  $\text{WO}_3$ . On the other hand, by increasing the  $\text{TiO}_2$  content, the total band gap of the photocatalyst decreases [42].

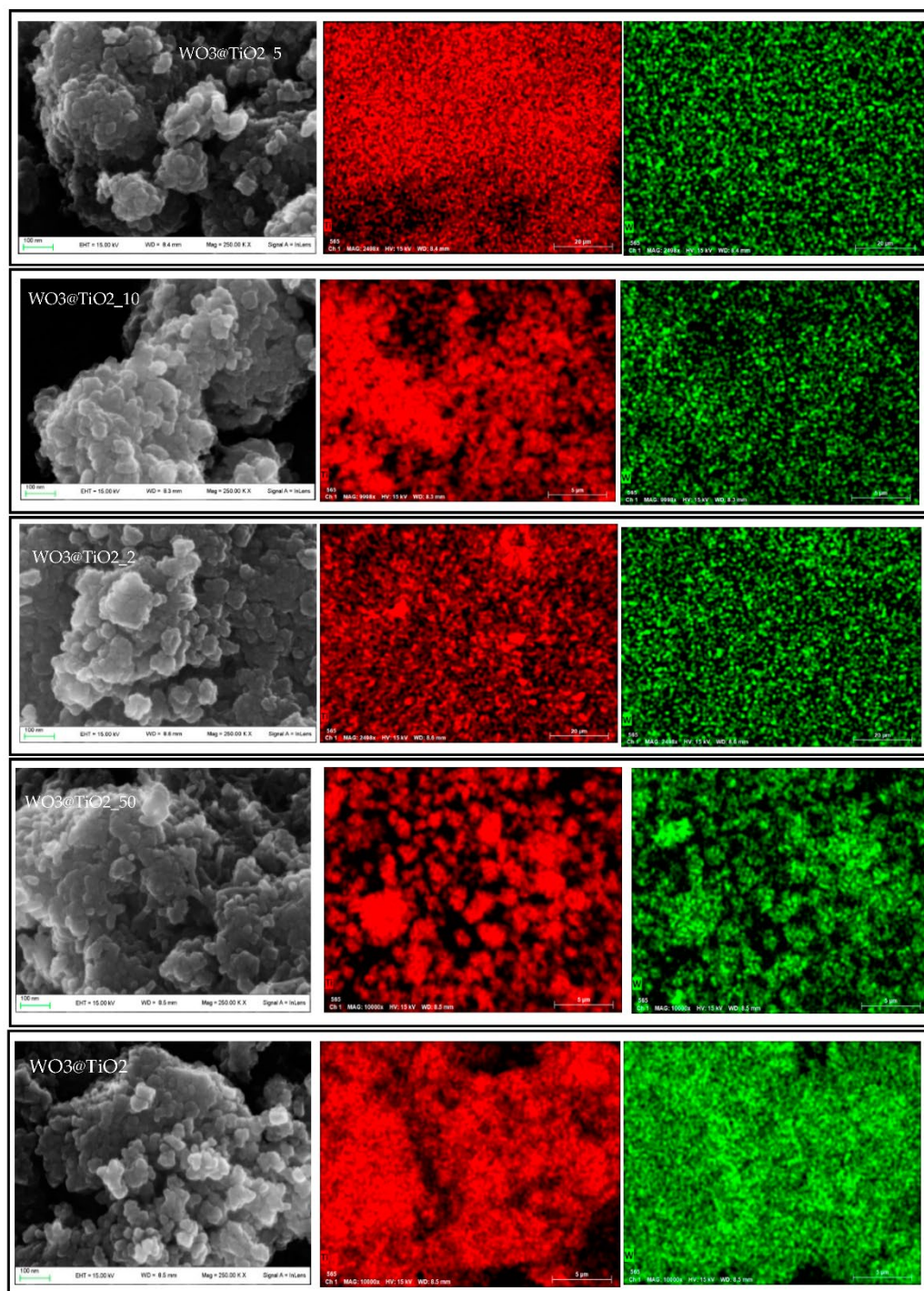
According to the shape of nitrogen adsorption–desorption isotherms reported in Figure 5 and the IUPAC classification [43], all the photocatalysts calcined at 400 °C can be classified as mesoporous materials of type IV, as confirmed by the values of  $C_{\text{BET}}$  in Table 1, containing other quantitative data.



**Figure 5.** Nitrogen adsorption–desorption isotherms for the synthesized samples.

Figure 6 displays the SEM images and the elemental mapping of Ti and W for the synthesized composites, whereas the EDX spectra are reported in the Supplementary Information (Table S1).

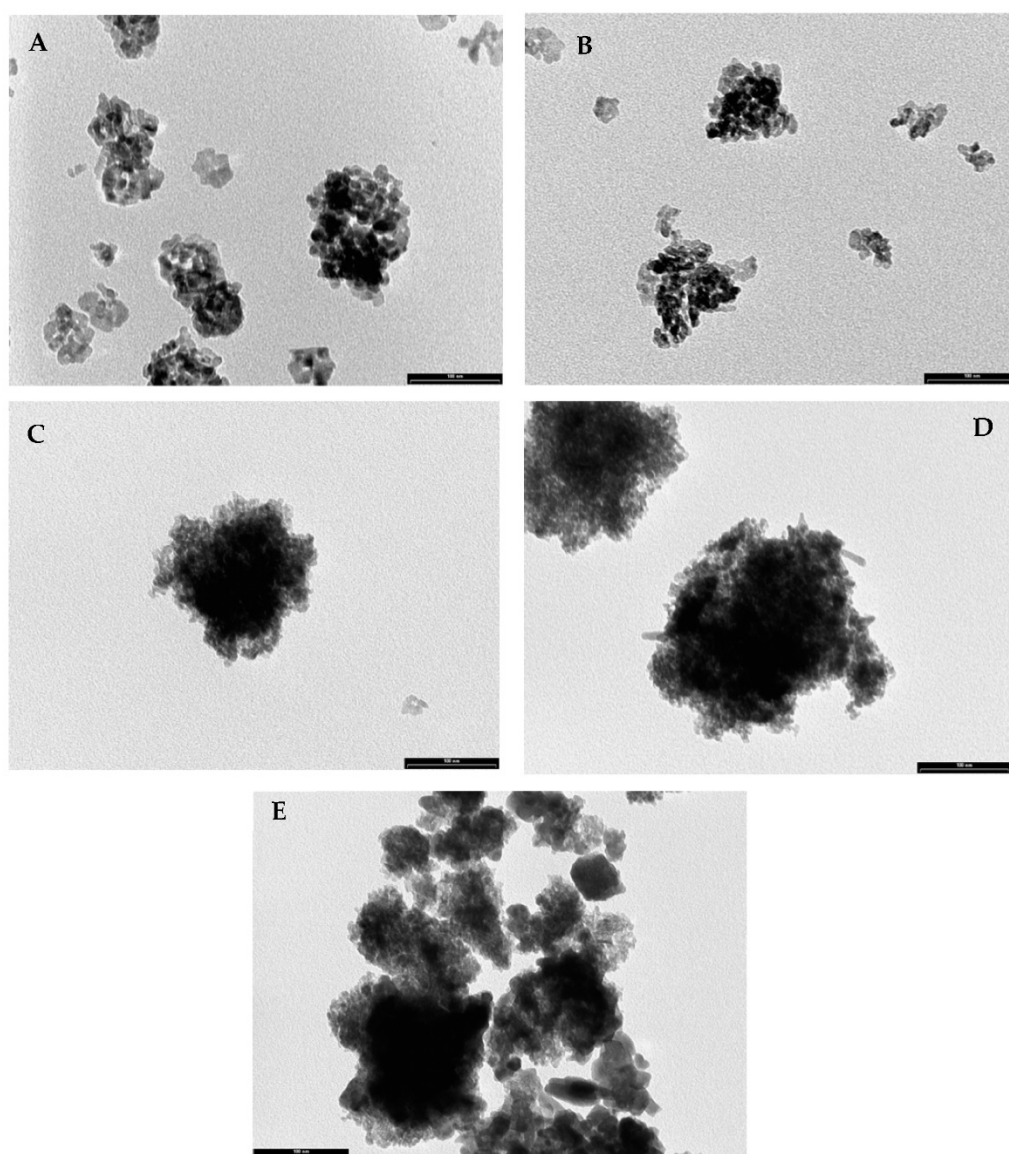




**Figure 6.** SEM images and elemental mapping of the synthesized materials. (Left) sample analyzed, (middle, red-colored) titanium map, (right, green-colored) tungsten map.

The elemental maps of W and Ti by X-ray energy dispersion (EDX) in the composites demonstrate that  $\text{TiO}_2$  and  $\text{WO}_3$  are well dispersed in each material. All the  $\text{WO}_3@ \text{TiO}_2$  show a globular-like morphology with particle sizes ranging from 60 to 5 nm. The particles are aggregated by sharing corners or edges that probably involve the formation of Ti–O–W bonds [44]. The same information was obtained by TEM investigations (Figure 7) showing nanoparticles of 12–35 nm that gradually aggregate with the  $\text{WO}_3$  load, reaching up to 60 nm in size.



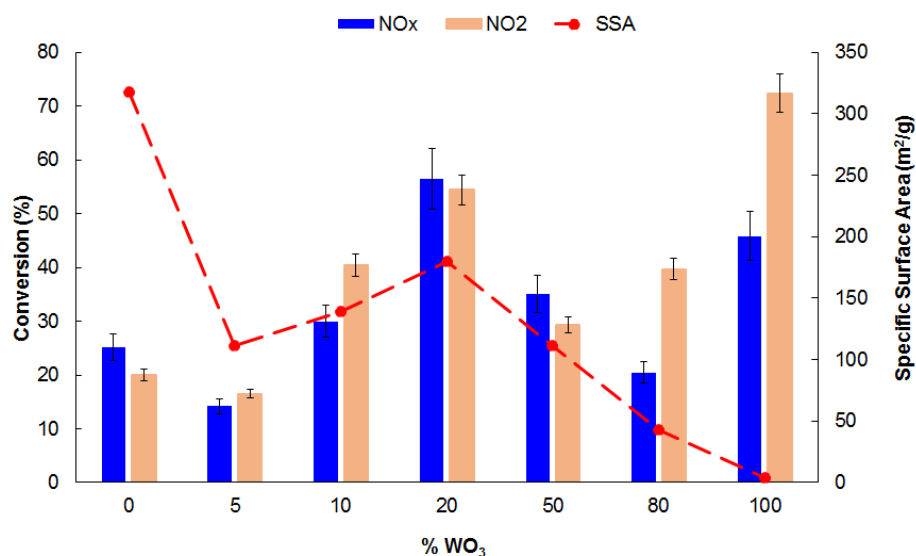


**Figure 7.** TEM images of  $\text{WO}_3@\text{TiO}_2_5$  (A),  $\text{WO}_3@\text{TiO}_2_{10}$  (B),  $\text{WO}_3@\text{TiO}_2_{20}$  (C),  $\text{WO}_3@\text{TiO}_2_{50}$  (D),  $\text{WO}_3@\text{TiO}_2_{80}$  (E).

## 2.2. Photocatalytic and Biological Activities of Catalysts

### 2.2.1. $\text{NO}_x$ Photocatalytic Degradation

The photocatalytic activity of the  $\text{TiO}_2/\text{WO}_3$  composites, as well as single-phase photocatalysts, referred to both  $\text{NO}_x$  ( $\text{NO} + \text{NO}_2$ ) and  $\text{NO}_2$  conversion under visible light irradiation (Figure 8).



**Figure 8.** Dependence of NO<sub>x</sub> and NO<sub>2</sub> photodegradation on WO<sub>3</sub> content and specific surface area (SSA) of the materials.

According to the literature [22,31,45], for this type of material the photocatalytic degradation of NO<sub>x</sub> consists of a photo-oxidation process, where both NO and NO<sub>2</sub> species are first adsorbed on the surface of the heterostructures and then converted into the corresponding oxidation product (NO<sub>3</sub><sup>-</sup>) under light irradiation. Although the formation of the oxidation products was well documented in the literature [46,47], an analytical confirmation was not performed in the present work due to the low NO<sub>x</sub> concentrations used during the test.

For all the experiments, the initial concentration of NO<sub>x</sub> was approximately 500 ppb. The NO<sub>x</sub> photodegradation results are summarized in Figure 8.

As expected, despite its extraordinarily high surface area, TiO<sub>2</sub> shows poor activity towards NO<sub>x</sub> conversion under visible light irradiation, leading to NO<sub>x</sub> and NO<sub>2</sub> degradation in 25% and 20%, respectively. On the other hand, according to the band gap value, regardless of its low surface area, pristine WO<sub>3</sub> exhibits a good photoactivity towards NO<sub>2</sub> photodegradation (72%) in 3 h, whereas the NO<sub>x</sub> abatement is only 46%. In fact, as reported in the scientific literature [19–22], WO<sub>3</sub> can remarkably reduce NO<sub>2</sub> into NO in the presence of N<sub>2</sub>. This is confirmed by the results reported in Figure S3, showing for the WO<sub>3</sub> sample an increase in the NO concentration during the reaction. This makes pristine WO<sub>3</sub> not efficient in the NO<sub>x</sub> abatement, because, as it is known, in air NO is immediately reoxidized to NO<sub>2</sub>. In this regard, from the pioneering investigations of Yu et al., carried out under nitrogen atmosphere and UV irradiation, a 20% conversion of NO<sub>2</sub> into NO can be inferred [20]. The present results demonstrate that even under visible light irradiation, the percentage of NO formation from NO<sub>2</sub> is of the same order (24%), calculated by Equation (1):

$$\text{NO produced (\%)} = \frac{[\text{NOt}] - [\text{NOi}]}{[\text{NOt}]} \quad (1)$$

where [NOt] is the NO concentration at the end of the reaction (after 3 h of light irradiation), [NOi] is the NO concentration before light irradiation.

If compared to the single-phase photocatalysts (TiO<sub>2</sub> and WO<sub>3</sub>), the photoactivity of WO<sub>3</sub>/TiO<sub>2</sub> heterostructures strongly depends on their composition. More in detail, the activity of the catalysts gradually increases with the WO<sub>3</sub> load, reaching the highest photodegradation efficiency by WO<sub>3</sub>@TiO<sub>2</sub>\_20 (54.4% NO<sub>x</sub> conversion and 56.4% NO<sub>2</sub> abatement), whereas it decreases for a percentage of WO<sub>3</sub> > 20. These results are in line

with the pioneering investigations of Balayeva et al., who tested the photocatalytic activity of TiO<sub>2</sub>/WO<sub>3</sub> composites towards NO degradation under UV irradiation, obtaining a ca. 35% of conversion for heterostructures characterized by a 1% and 2.5% of WO<sub>3</sub> load [48].

For a very high amount of WO<sub>3</sub> (WO<sub>3</sub>@TiO<sub>2</sub>\_80), these latter composites maintain the photoactive capability of pristine WO<sub>3</sub>, converting NO<sub>2</sub> to NO. The different photoactivity of the materials may be due to a combination of factors.

First of all, it can be assumed that for a WO<sub>3</sub> load < 20%, TiO<sub>2</sub> and WO<sub>3</sub> only play their own photocatalytic role, and coupled photocatalysts are not formed. In this case, the low activity of TiO<sub>2</sub> prevails because it is the major component. In contrast, for a large amount of WO<sub>3</sub>, the fast e<sup>-</sup>/h<sup>+</sup> recombination of the WO<sub>3</sub> component predominates.

On the contrary, the absence of WO<sub>3</sub> peaks in the XRD spectra of the WO<sub>3</sub>@TiO<sub>2</sub>\_20 sample suggests that its increased photoactivity is not related to the formation of crystalline tungsten oxide but is probably due to the presence of WO<sub>3</sub> centers on the surface of TiO<sub>2</sub> acting as electrons/holes separators [49]. When the test was carried out using WO<sub>3</sub>@TiO<sub>2</sub>\_20\*, in order to observe the effect of calcination temperature on the photoactivity of the material, the percentage of NO<sub>x</sub> and NO<sub>2</sub> degradation dropped to ca. 20%, confirming that the thermal treatment acts by reducing the WO<sub>3</sub> content in the WO<sub>3</sub>@TiO<sub>2</sub> heterostructure and as a consequence of its activity.

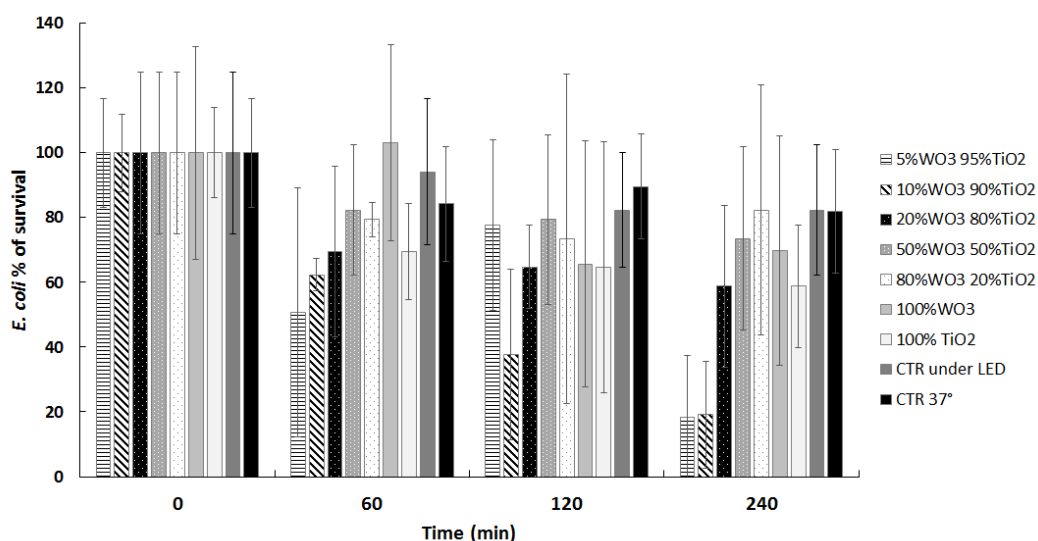
It is known that the quantum efficiency of photocatalytic reactions carried out by heterogeneous photocatalysts depends on the competition between the recombination of photogenerated electrons and holes and the transfer of both electrons and holes at the interface of the material. Extending the electrons and holes recombination time and increasing the transfer rate of electrons at the interface enhance the quantum efficiency positively. As reported in the literature [31], the formation of TiO<sub>2</sub>/WO<sub>3</sub> heterostructures leads to enhanced charge carrier lifetimes, due to the transfer of photogenerated electrons in the TiO<sub>2</sub> to WO<sub>3</sub> CB, and at the same time to the entrapment of the photogenerated holes within the TiO<sub>2</sub> particle. Both these phenomena make charge separation more efficient.

Finally, the effect related to the different surface area values cannot be ignored. As for the photocatalytic activity, the surface area values also seem to be correlated to the WO<sub>3</sub> load and the most active catalyst (WO<sub>3</sub>@TiO<sub>2</sub>\_20) is also the one with the highest surface area (Figure 8, Table 1). At first glance, the results obtained by the WO<sub>3</sub>@TiO<sub>2</sub>\_20 photocatalyst seem to contrast with those of Mendoza et al. [22], who report very high NO conversion values under visible light irradiation in similar conditions. The different photocatalytic activity of the WO<sub>3</sub>@TiO<sub>2</sub>\_20 sample compared to those reported by Mendoza and coworkers can be reasonably attributed to the real WO<sub>3</sub> content in the synthesized composites that it is not specified in the work of the author [22].

#### 2.2.2. E. coli Photoinactivation

The antibacterial activity of WO<sub>3</sub>@TiO<sub>2</sub> composites, as well as single-phase photocatalysts, was assessed by determining the percentage of *E. coli* cell survival following exposure to visible light. According to standard methods (ASTM E2149, 2001), values of survival ≤ 90% indicate the antibacterial activity of a given photocatalytic film.

As shown in Figure 9, none of the tested samples displays antibacterial activity under visible light irradiation, in agreement with what was reported in the literature for the WO<sub>3</sub>/TiO<sub>2</sub> catalyst [36].



**Figure 9.** Percentage of *E. coli* cell survival after exposure to visible light in the presence of the synthesized heterostructures.

Different catalysts (e.g., sulfur-doped carbon quantum dots loaded hollow tubular g-C<sub>3</sub>N<sub>4</sub>) give degradation of *E. coli* cells instead, under visible light [50].

Based on the morphology of nanoparticles (Figure 7), we speculate that the nanoparticle aggregation of TiO<sub>2</sub>/WO<sub>3</sub> hinders a suitable surficial interaction with the bacteria and the catalyst cytotoxicity.

### 3. Materials and Methods

#### 3.1. Chemicals

Tungstic acid (H<sub>2</sub>WO<sub>4</sub>, 99% Merck), AMT 100 TiO<sub>2</sub> (Tayca Corporation, WP0097, Osaka, Japan), ammonium hydroxide solution (ACS reagent, 28.0–30.0% Merck & Co., St. Louis, MO, USA), hydrochloric acid (HCl 36%, Suprapur®, Supelco, Belfont, PA, USA) were used as received.

#### 3.2. Synthesis of TiO<sub>2</sub>/WO<sub>3</sub> Series

To synthesize 1 g of TiO<sub>2</sub>/WO<sub>3</sub> composite, a proper amount of TiO<sub>2</sub> was dispersed in 25 mL of 2 M ammonium hydroxide solution under constant stirring (solution A). The WO<sub>3</sub> precursor solution was prepared by dissolving a proper amount of tungstic acid H<sub>2</sub>WO<sub>4</sub> in 25 mL of 2 M ammonium hydroxide solution under constant stirring (solution B). The quantity of TiO<sub>2</sub> and H<sub>2</sub>WO<sub>4</sub> used is reported in Table 2. The two solutions were stirred for 30 min at room temperature, then solution B was added to solution A and the stirring was continued for another 2 h at room temperature. Then, the solvent was evaporated, heating the mixture at 110 °C. The white-yellow powder was treated with 0.5 M hydrochloric acid solution and dried again. The final powder was washed with de-ionized water abundantly, dried at 100 °C overnight, and calcinated in the air at 400 °C for 2 h (heating rate 8 °C·min<sup>-1</sup>). The synthesized samples containing different TiO<sub>2</sub>/WO<sub>3</sub> w/w ratios (95:5, 90:10, 80:20, 20:80) were properly characterized and tested for NO<sub>x</sub> photodegradation under visible light irradiation.

**Table 2.** Labels of the WO<sub>3</sub>/TiO<sub>2</sub> composites, WO<sub>3</sub>/TiO<sub>2</sub> w/w ratio, calcination temperature, and reaction yield.

Label	TiO <sub>2</sub> (g)	H <sub>2</sub> WO <sub>4</sub> (g)	WO <sub>3</sub> /TiO <sub>2</sub> (w/w Ratio)	Calcination Temperature (°C)	Yield (%)
WO <sub>3</sub>	0.00	1.08	100:0	400	95.2
WO <sub>3</sub> @TiO <sub>2</sub> _80	0.20	0.86	80:20	400	95.7
WO <sub>3</sub> @TiO <sub>2</sub> _50	0.50	0.55	50:50	400	95.1
WO <sub>3</sub> @TiO <sub>2</sub> _20	0.80	0.22	20:80	400	94.8
WO <sub>3</sub> @TiO <sub>2</sub> _10	0.90	0.11	10:90	400	96.2
WO <sub>3</sub> @TiO <sub>2</sub> _5	0.95	0.05	5:95	400	95.7
WO <sub>3</sub> @TiO <sub>2</sub> _20*	0.80	0.22	20:80	600	60.9
TiO <sub>2</sub>	1.00	0.00	0:100	400	98.5

WO<sub>3</sub> was synthesized by the same procedure using a TiO<sub>2</sub>-free solution A.

An aliquot of the TiO<sub>2</sub>/WO<sub>3</sub> composite with an 80:20 w/w ratio was calcined at 600 °C.

Table 2 reports a list of synthesized composites with the corresponding label, WO<sub>3</sub>/TiO<sub>2</sub> w/w ratio, calcination temperature, and reaction yield, calculated by Equation (2):

$$\text{Yield (\%)} = \frac{g(\text{H}_2\text{WO}_4) \times \text{molar mass}(\text{WO}_3) + g(\text{TiO}_2)}{\text{molar mass}(\text{H}_2\text{WO}_4)} \times 100 \quad (2)$$

### 3.3. Characterization Methods

X-Ray Diffraction (XRD) measurements investigated the crystalline structure on a PW3830/3020 X'Pert diffractometer from PANalytical working Bragg–Brentano, using the Cu Kα1 radiation (k = 1.5406 Å).

FT-IR spectra were recorded in the range of 400–4000 cm<sup>-1</sup> with a resolution of 0.5 cm<sup>-1</sup> by an FT-IR 410 spectrophotometer dispersing a few milligrams of each material in anhydrous KBr. The morphology of the catalysts was inspected employing high-resolution electron transmission microscopy (HR-TEM), using a JEOL 3010-UHR instrument (acceleration potential: 300 kV; LaB6 filament), and by scanning electron microscopy (SEM), using a Zeiss LEO 1525 field emission microscope. The samples were “dry” dispersed on lacey carbon Cu grids for TEM analyses, whereas SEM analyses were carried out without any pre-treatment of the samples.

For the band gap determinations, diffuse reflectance spectra of the powders were collected on a UV–Vis diffuse reflectance spectra using a scanning spectrophotometer PerkinElmer, Lambda 35, which was equipped with a diffuse reflectance accessory. A thin film of each sample was placed in the sample holder on an integrated sphere for the reflectance measurements. A KBr pill was used as the reference material. Data were elaborated using the Kubelka–Munk function (Equation (3)), which expresses the absorbance as a function of reflectance (F(R)) [51]:

$$F(R) = (1 - R)^2/2R \quad (3)$$

where R = reflectance of the powder.

The band gap values were determined by performing the first derivative of the Kubelka–Munk function (Equation (4)):

$$dF(R)/d\lambda \quad (4)$$

where λ = wavelength of the incident radiation. The energy of the radiation at which the first derivative dF(R)/dλ shows the maximum was taken to estimate the band gap values.



Specific surface area and porosity distribution were determined by processing N<sub>2</sub> adsorption–desorption isotherms at 77 K (Micromeritics Tristar II 3020 (Micromeritics)) with Brunauer–Emmett–Teller (BET) and Barrett–Joyner–Halenda analyses. Before the analysis, samples were heat-treated (T = 150 °C, 4 h, N<sub>2</sub>) to remove adsorbed foreign species.

### 3.4. NO<sub>x</sub> Photodegradation Tests

A photocatalytic film of each sample was deposited by drop-casting on glass supports as follows: a suspension of 0.050 ± 0.001 g of photocatalyst in 5 mL of isopropanol was deposited on a glass plate (230 × 19 mm). Once the solvent was evaporated, the photocatalyst was placed inside a 20 L Pyrex glass cylindrical batch reactor for the photocatalytic tests. The photocatalytic tests were performed by a mixture of NO and NO<sub>2</sub> in air. The starting inlet gas contains only NO<sub>2</sub>, but the chemical equilibrium between NO and NO<sub>2</sub> is established as it is exposed to air. An LED lamp (350 mA, 9–48 V DC, 16.8 W) with emissions in the 400–700 nm range was used as the light source. The luminous intensity (lux) was measured using an illumination meter (Delta Ohm photo/radiometer HD 2102.2) and was 2900 lx to estimate the light intensity. It was then converted to the irradiance unit (in mW/cm<sup>2</sup>) [19], obtaining a light intensity of 3.24 mW/cm<sup>2</sup>.

The NO<sub>x</sub> initial concentration was 500 ± 50 ppb. A chemiluminescence analyzer measured the NO<sub>x</sub> concentration after 30, 60, and 180 min of exposure to light irradiation (ENVEA AC32e).

### 3.5. Antibacterial Assay

Cultures of *E. coli* MG1655 [52] were grown at 37 °C in Luria–Bertani (LB) medium (10 g/L tryptone, 5 g/L yeast extract, 10 g/L NaCl) or LB-agar medium (LB medium with 10 g/L agar). Bacteria (200 mL) were collected by 10 min centrifugation at 5000 rpm, washed in PBS 1X (Merck & Co., St. Louis, MO, USA), and resuspended in the same volume of PBS 1X. Stationary phase cultures of *E. coli* were diluted up to optical density at 600 nm (OD<sub>600</sub>) of 0.05 and then grown aerated up to 0.6.

Films of the synthesized heterostructures were prepared as follows. A total of 100 mg of each material was dispersed in 8 mL of isopropanol and deposited on a Petri dish (90 mm in diameter) by drop-casting and air-dried.

10 mL of bacterial cells were added to Petri dishes. The plates were irradiated with visible light (2900 lux obtained with an LED lamp) and removed from light at different time points (60, 120, and 240 min). As controls, bacterial cells were deposited onto empty Petri dishes and irradiated (CTR under LED) or incubated at 37 °C under dark (CTR). Viable bacteria expressed as CFU/mL (colony-forming unit /mL) were enumerated at t = 0, 60, 120, and 240 min by plating suitable dilutions onto LB-agar plates following incubation at 37 °C for 18 h.

The percentage of bacterial survival is expressed as follows:

(average of viable bacteria at a given time/average of viable bacteria CTR 37° at t = 0 min) × 100.

The average of viable bacteria is calculated from at least three independent experiments. Values of survival ≤90% indicate the antibacterial activity of a given photocatalytic film.

## 4. Conclusions

In this study, we investigated the photoactivity of WO<sub>3</sub> and TiO<sub>2</sub>/WO<sub>3</sub> composites towards NO<sub>2</sub> degradation under visible light, in order to clarify the numerous conflicting data reported in the literature so far. It was demonstrated that the photoactivity of TiO<sub>2</sub>/WO<sub>3</sub> heterostructures are strongly related to their composition. For WO<sub>3</sub>@TiO<sub>2</sub> materials characterized by low tungsten trioxide content (<20%), TiO<sub>2</sub> and WO<sub>3</sub> are present

as separate phases, each playing their own photocatalytic role, whereas coupled photocatalysts are not formed.

The composite with a  $\text{WO}_3$  load of 20% was the most efficient photocatalyst, extending the electrons and holes recombination time and promoting the transfer rate of electrons at the interface. The high activity of the material can be explained with its high surface area value and with the presence of  $\text{WO}_3$  centers on the surface of  $\text{TiO}_2$  acting as electrons/holes separators. However, if the  $\text{WO}_3$  load is higher than 20%, a fast  $e^-/h^+$  recombination can occur and the ability of tungsten trioxide to reduce  $\text{NO}_2$  to  $\text{NO}$  prevails over the composites' capability to photo-oxidize  $\text{NO}_2$  to  $\text{NO}_3^-$ . Moreover, the photodegradation activity of the heterostructures can be attributed to the oxidizing effect of holes. Moreover, it was demonstrated that high-temperature calcination leads to a partial sublimation of the  $\text{WO}_3$  component that causes a decrease in heterostructure activity. As for the lack of the bacteria degradation, we tentatively suggest that the aggregation of nanoparticles hinders an efficient surface contact between bacteria and catalyst.

**Supplementary Materials:** The following supporting information can be downloaded at: [www.mdpi.com/article/10.3390/catal12080822/s1](http://www.mdpi.com/article/10.3390/catal12080822/s1): Table S1: EDS analysis, Figure S1: EDX spectrum of (A)  $\text{WO}_3@/\text{TiO}_2_{20}^*$  and (B)  $\text{WO}_3@/\text{TiO}_2_{20}$ , Figure S2: Tauc plot of  $\text{TiO}_2$  (A),  $\text{WO}_3@/\text{TiO}_2_{5}$  (B),  $\text{WO}_3@/\text{TiO}_2_{10}$  (C),  $\text{WO}_3@/\text{TiO}_2_{20}$  (D),  $\text{WO}_3@/\text{TiO}_2_{50}$  (E),  $\text{WO}_3@/\text{TiO}_2_{80}$  (F),  $\text{WO}_3$  (G), Figure S3: Dependence of  $\text{WO}_3$  content (%) in the composites versus and  $\text{NO}$  production.

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